## MATRIX INDUCED DIFFERENCES IN LUMINESCENCE PROPERTIES OF LANTHANIDE-SUBSTITUTED MIXED-METAL L: $Y_{3} A L L_{5-x} M_{x} O_{12}(M=I n, C r, L=C e, E u, E r A N D ~ T b ; ~ 0.50<x<2.25)$ GARNETS SYNTHESIZED BY SOL-GEL METHOD

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## Introduction

Yttrium aluminum garnet (YAG) shows exceptional chemical stability and therefore doped or substituted with $\mathrm{Ce}^{3+}$, $\mathrm{Eu}^{3+}$,
$\mathrm{Tb}^{3+}, \mathrm{Cr}^{3+}, \mathrm{Sm}^{3+}$, $\mathrm{Dy}^{3+}$ or $\mathrm{Tm}^{3+}$ is employed as the host material of multicolored phosphors. By selecting corresponding lanthanide ions may be produced red, green and blue (RGB) emission for use in tricolor white light $[1-3]$
It is known that the chemical composition of host material influences of optical properties of such phosphors considerably [4]. The matrixes of garnets could be modified by replacing different molar part of aluminium or gallium by other metals in yttrium aluminium or yttrium gallium garnets (YAG, YGG). Therefore, in the present work the sinterability, microstructural and luminescence properties evolution of lanthanide-doped and lanthanide-substituted mixed-metal $Y_{3} \mathrm{~A}_{5}$. $x^{M} \mathrm{O}_{\mathrm{O}} \mathrm{O}_{12}:$ L ( $\mathrm{L}=\mathrm{Ce}, \mathrm{Eu}, \mathrm{Er}$ or $\mathrm{Tb} ; \mathrm{M}=\mathrm{In}, \mathrm{Cr}, 0.75 \leq \mathrm{x} \leq 2.0$ ) garnets powders synthesized by an aqueous sol-gel process were investigated.

## Experimental

The gels were prepared using stoichiometric amounts of analytical-grade $\mathrm{Y}_{2} \mathrm{O}_{3}, \mathrm{Al}\left(\mathrm{NO}_{3}\right)_{3} \cdot 9 \mathrm{H}_{2} \mathrm{O}, \mathrm{In}_{2} \mathrm{O}_{3}$ or $\mathrm{Cr}\left(\mathrm{NO}_{3}\right)_{3} \cdot 9 \mathrm{H}_{2} \mathrm{O}$ and $\mathrm{Ce}\left(\mathrm{NO}_{3}\right)_{3}, \mathrm{Eu}_{2} \mathrm{O}_{3}, \mathrm{Er}_{2} \mathrm{O}_{3}$ and $\mathrm{Tb}_{4} \mathrm{O}_{7}$ as starting materials and 1,2-ethanediol as complexing agent. The oven dried ( $100^{\circ} \mathrm{C}$ in air) gel powders were ground in an agate mortar and preheated for 2 h at $800^{\circ} \mathrm{C}$ in air. After an intermediate grinding in an agate mortar the powders were additionally sintered for 10 h at $1000{ }^{\circ} \mathrm{C}$.
The synthesized samples were characterized by X -ray diffraction analysis, UV-visible and emission spectra.

## Conclusions

We have demonstrated that reflectance, exitation and luminescence spectra intensity of $\mathrm{L}: \mathrm{Y}_{3} \mathrm{Al}_{5-\mathrm{x}} \mathrm{In}_{\mathrm{x}} \mathrm{O}_{12}(\mathrm{~L}=\mathrm{Ce}, \mathrm{Eu}, \mathrm{Er}$ or b) is slighty dependant on the phase purity. Though, wavelength of peaks is almost constant. When $3 \%$ yttrium is substituted by cerium in yttrium-aluminium-indium the samples have one peak with wavelength maximum at 530 nm due to $[\mathrm{Xe}] \mathrm{d}^{5} \rightarrow[\mathrm{Xe}]^{f^{1}}$ transition. When yttrium is substituted by erbium, europium or terbium each sample contains several emission peaks. $\mathrm{Y}_{3} \mathrm{Al}_{5-\mathrm{x}} \mathrm{In}_{\mathrm{x}} \mathrm{O}_{12}: 3 \% \mathrm{Eu}^{3+}$ samples have several peaks due to the ${ }^{5} \mathrm{D}_{0} \rightarrow{ }^{7} \mathrm{~F}_{\mathrm{j}}(\mathrm{j}=0,1,2,3,4)$ transition between the wavelength of $580-720 \mathrm{~nm}$. Meanwhile $\mathrm{Y}_{3} \mathrm{Al}_{5-\mathrm{x}} \mathrm{In}_{\mathrm{x}} \mathrm{O}_{12}: 3 \% \mathrm{Er}{ }^{3+}$ samples contains several peaks due to the ${ }^{4} \mathrm{~S}_{3 / 2} \rightarrow{ }^{4} \mathrm{I}_{15 / 2}$ transition between the wavelength of $520-570 \mathrm{~nm}$ and ${ }^{4} \mathrm{~F}_{9 / 2} \rightarrow{ }^{4} \mathrm{I}_{15 / 2}$ transition between the wavelenth $650-680 \mathrm{~nm}$. The peaks of $\mathrm{Y}_{3} \mathrm{Al}_{5-\mathrm{x}} \mathrm{In}_{x} \mathrm{O}_{12}: 3 \% \mathrm{~Tb}^{3+}$ samples are between the wide wavelength of $370-700$ nm due to the ${ }^{5} \mathrm{D}_{3} \rightarrow{ }^{7} \mathrm{~F}_{\mathrm{j}}(\mathrm{j}=6,5,4)$ and ${ }^{5} \mathrm{D}_{4} \rightarrow{ }^{7} \mathrm{~F}_{\mathrm{j}}(\mathrm{j}=5,4,3,2)$ transitions. The emission spectra of $\mathrm{Y}_{3} \mathrm{Al}_{5-\mathrm{x}} \mathrm{In}_{\mathrm{x}} \mathrm{O}_{12}: \mathrm{L}(\mathrm{L}=$ $\mathrm{Ce}, \mathrm{Eu}, \mathrm{Er}, \mathrm{Tb}$ ) is nearly related with emission spectra of YAG:L ( $L=C e, E u, E r, T b$ ).
The tentative investigations showed that the modification of matrix by change aluminium with chromium dramatically has influenced the luminescence properties. $\mathrm{Y}_{3} \mathrm{Al}_{4.25} \mathrm{Cr}_{0.75} \mathrm{O}_{12}: 3 \% \mathrm{Eu}^{3+}$ is optically inactive material and does not show any emission

Results



## References

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